



## Electroactivity and electrochemical stability of Graphene-PANI nanocomposites for supercapacitor electrodes

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### ABSTRACT

In this paper, a nanocomposite of reduced graphene oxide and polyaniline (rGO/PANI) was prepared by in-situ method, which can be directly used as electrodes materials for a flexible supercapacitor. That rGO was dispersed on PANI supporter created a rGO/PANI nanocomposite which has high surface area. At the same current density, the specific capacitance of rGO/PANI is  $1121.2 \text{ F g}^{-1}$ , while that of rGO is  $182.7 \text{ F g}^{-1}$  at  $1 \text{ A g}^{-1}$  in  $\text{H}_2\text{SO}_4$  1 M medium. After 3000 cycles working at  $5 \text{ A g}^{-1}$ , the specific capacitance retention of rGO/PANI is about 75 % compared to the first cycle, higher than rGO (62.7 %). The graphene and conductive polymer combination, which formed a high electroactivity and electrochemical stability nanocomposite, can be applied on coating material of supercapacitor electrode.

### Introduction

Supercapacitor is a device that is capable of fast charging and discharging, which has much higher energy storage than conventional capacitors. To meet the advanced technical requirements in manufacturing supercapacitors, the spongy materials with high surface area were usually used to coat on the electrode surface. Among those materials, owing to its high specific surface area and excellent conductivity, graphene has been drawing tremendous attention.

The specific capacitance values of the supercapacitors with electrodes coated by graphene-based materials, published in the studies, range from 100 to nearly  $2000 \text{ F g}^{-1}$  [1-5]. In general, there are two ways to use graphene in electrode fabrication of supercapacitor: first, graphene material as electrode for dual-layer electrostatic deposition supercapacitor [1, 2]; second,

graphene as a part of combination with metal oxides [3, 4] or conductive polymers used as electrode for fake electrochemical capacitor supercapacitors [5, 6].

For the first approach, graphene oxide is mainly used as an intermediate. However, the conductivity of the reduced graphene oxide needs to be improved and the number of layers of graphene should be reduced in order to increase the specific surface area. In the second approach, the graphene material combined with metal oxides, such as  $\text{MnO}_2$  [7, 8] and  $\text{RuO}_2$  [9, 10], or the graphene-polymers complexes, which have high specific capacitance values, improves the conductivity of the graphene, and at the same time results in the combination of graphite-based dual-layer electrostatic deposition and redox reaction. Whereas graphene-metal oxide composite material should be studied to improve the ability to disperse metal oxide on the surface of graphene and develop the overall

surface area of the graphene-polymer material, the graphene-polymer material system not only owns a large surface area, but also enhances the durability of the material during charging and discharging when supercapacitor is working.

This paper presents the results of the investigations of the electroactivity of graphene-PANI as well as the stability during the charge-discharge of this material in electroplating applications of supercapacitor.

## Experimental

All chemicals were purchased from Sigma Aldrich. Deionized water was used throughout the study. Graphene oxide (GO) was synthesized by the modified Hummers method as presented in [11]. For the synthesis of reduced graphene oxide (rGO), GO was dispersed in ethylene glycol (EG) solution and stirred at 120 °C for 24 h. Afterwards, PANI was added to solution of 1.5 g rGO in 100 mL HCl 0,5 M, with ratio  $m_{\text{PANI}}/m_{\text{rGO}} = 1/2$ , the mixture was stored at 5 °C. Then, 1.25 g ammoni pesulfat (pre-cooled 5 °C) was poured into the above mixture, and the resultant mixture was left at 5 °C for 5 h. The obtained nanocomposite foam was filtered, washed until pH = 7. Then it was dispersed in a mixture containing 0,9 mL deionized water and 0,3 mL Nafion 1 % under ultrasonication for 30 min.

Powder XRD patterns of samples were recorded with a D8 Advance diffractometer (Bruker). The morphology and microstructure of samples were investigated with a transmission electron microscope (TEM) JEOL JEM 2010 equipped operated at 200 kV. Surface area and pore size distribution of the samples were obtained by the BET measurement in the ASAP 2010 nitrogen adsorption equipment. Electrochemical measurements were performed on a CPA-ioc-HH5B Polarographic Analyser with a conventional three-electrodes cell. The working electrode is a glassy carbon electrode (5 mm in diameter), which was coated with 10  $\mu\text{L}$  of the above catalyst ink and dried at room temperature. The counter electrode is a platinum electrode and the reference electrode is an Ag/AgCl. The electrochemical solution ( $\text{H}_2\text{SO}_4$  1M) was deaerated by pure nitrogen gas prior to measurements.

The electrochemical operability of the material was evaluated by the specific capacitance ( $C_s$ ) [4, 10], using the following equation:

$$C_s = \frac{dQ}{dV} = \frac{idt}{dV} = \frac{i}{\nu} \quad (1)$$

Where,  $C_s$  is the specific capacitance ( $\text{F g}^{-1}$ ),  $Q$  is the charge stored by each electrode ( $\text{C g}^{-1}$ ),  $V$  is the potential (V),  $i$  is the current density ( $\text{A g}^{-1}$ ) - calculated from CV curves,  $\nu$  is the potential scan rate ( $\text{V s}^{-1}$ ). The CV analysis was carried at various scan rates, ranging from 1 to 100  $\text{mV s}^{-1}$ , in a potential range of 0  $\div$  1 V.

$$C_s = \frac{I\Delta t}{m\Delta V} \quad (2)$$

Where,  $m$  is the mass of electroactive material (g),  $I$  is the current applied (A),  $\Delta t$  is the discharging time (s),  $\Delta V$  is the potential range (V). All these values were identified by the galvanostatic charge/discharge techniques in  $\text{H}_2\text{SO}_4$  1 M solution, in a potential range of 0  $\div$  1 V, at different current density 1, 2, 3, 4, 5  $\text{mA mg}^{-1}$ .

The Coulombic efficiency of material was calculated using the following equation [12]:

$$\eta = (t_d \cdot 100) / t_c \quad (\%) \quad (3)$$

Where,  $\eta$  is the Coulombic efficiency of material (%),  $t_d$  and  $t_c$  are the times of discharging and charging with the same current, respectively.

## Results and discussion

### Characteristic physico-chemical properties of the material

Fig. 1 shows the XRD patterns of rGO/PANI and rGO. From Fig. 1, it is realized that the two rGO/PANI and rGO materials have XRD patterns which are quite similar in shape, with the characteristic peak of graphene at  $2\theta \approx 25,14^\circ$ . Obviously, bearing rGO onto PANI causes no influence on the structure of graphene.

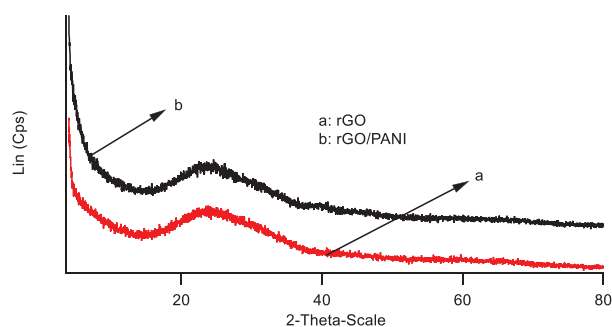


Fig 1. XRD patterns of rGO/PANI and rGO

Fig. 2 shows the TEM images of rGO (a) and rGO/PANI (b). In Fig. 2.b, in addition to graphene sheets as observed in Fig. 2.a, there are also adhesive layers between these graphene sheets. This is thought to be the polymer material in the crystalline structure of rGO/PANI.

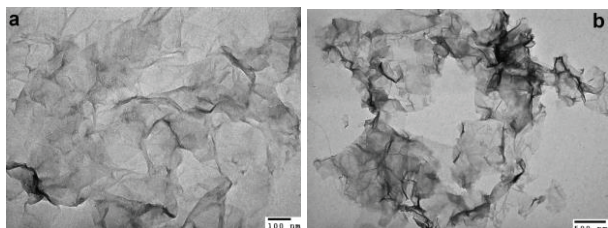


Fig 2. TEM images of rGO (a) and rGO/PANI (b)

The  $S_{\text{BET}}$  specific surface area measurement (Table 1) indicates that the rGO/PANI material has a much higher specific surface area than the rGO material, which is consistent with Li Tang's argument [13]. Accordingly, thin PANI layers covering the surface and alternating between graphene sheets increase the specific surface area of the material. This nearly double increase in rGO/PANI's specific surface area value over rGO is expected to improve the electrical storage, transmission and discharge properties of the material.

Table 1.  $S_{\text{BET}}$  value of rGO/PANI and rGO

| Sample   | The $S_{\text{BET}}$ specific surface area ( $\text{m}^2/\text{g}$ ) |
|----------|----------------------------------------------------------------------|
| rGO      | 257,42                                                               |
| rGO/PANI | 578,26                                                               |

#### Determining electroactive properties of rGO/PANI

Fig. 3 displays CV measurement result of the rGO/PANI. For ideal capacitors, the CV line is rectangular, highly symmetric because of low contact resistance [14]. For the three-electrode implant system, as in the study of large CV distortion, reduction of symmetry led to narrow loop and oblique angle [15]. At low scanning speed (less than  $50 \text{ mV s}^{-1}$ ), the CV lines obtained are nearly rectangular as in the case of ideal capacitors, resulting in simplicity and high accuracy in the determination of specific capacitance of rGO/PANI from the CV scan results. The specific capacitance values of rGO/PANI (calculated by formula (1)) reach  $82 \text{ F g}^{-1}$  and  $105 \text{ F g}^{-1}$  at scan rate of  $5 \text{ mV s}^{-1}$  and  $50 \text{ mV s}^{-1}$ , respectively. When the scan rate increases up to  $50 \text{ mV s}^{-1}$ , a faraday current is generated against the current direction that the material accumulates and releases in the sweeping current flow measurement, which affects the rate of change. the specific

capacitance as well as the reduction in accuracy when determining this value through the circulating current scanning line method. In general, the rGO/PANI's own capacitance increases with increasing scan rate.

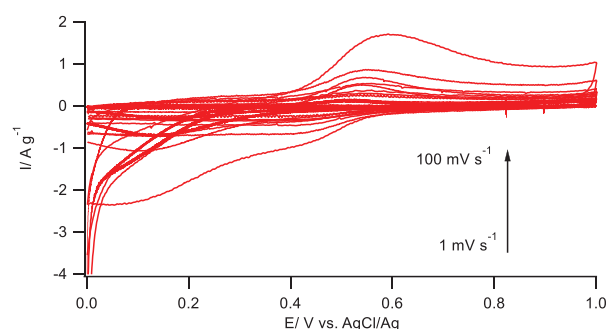


Fig 3. CV curves of rGO/PANI in  $\text{H}_2\text{SO}_4$  1 M at different scan rates.

The results of the charge-discharge cycle of rGO and rGO/PANI respectively are shown in Fig. 4, 5. From the results obtained, it can be easily to observe that total discharge time (of discharge cycle) is inversely proportional with the increase of the current density. According to formula (2),  $C_s$  is proportional to the discharge cycle, so it is inversely proportional to the current density. This is consistent with the results of previously published studies in the world [2-15]. In charge-discharge cycle, with fixed measuring range, the greater the current density is, the greater the amount of charge provided to the material in a unit of time is, which makes it easier to reach the voltage level, and shortens time of energy storage of the material ( $t_c$  decreases). It can also be understood that the lower the current density is, the more time required to fill or emit  $\text{H}^+$  ions is. With ideal capacitors, at the same constant current density, the charging cycle and the discharging cycle of the capacitor are equivalent, so the smaller  $t_c$  is, the smaller  $t_D$  is, which reduces  $C_s$ .

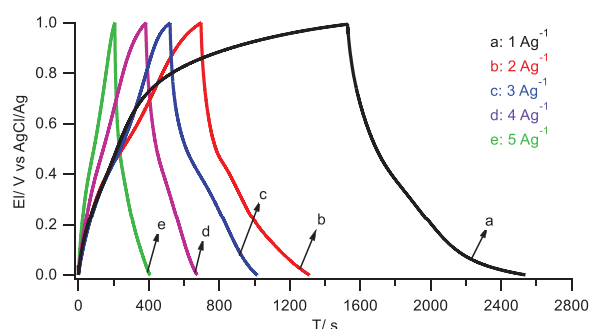


Fig 4. Galvanostatic charge/discharge curves of rGO/PANI in  $\text{H}_2\text{SO}_4$  1M at different current densities

Both rGO and rGO/PANI exhibited a very quickly increasing charge-discharge cycle when the current density decreased from  $5 \text{ A g}^{-1}$  to  $1 \text{ A g}^{-1}$ . Concurrently, the charge-discharge cycle of rGO/PANI was much higher than that of rGO, which means that the specific capacitance value of rGO/PANI was much greater than that of rGO, at any current density values. The specific capacitance values of the materials at the different current densities are listed in Table 2.

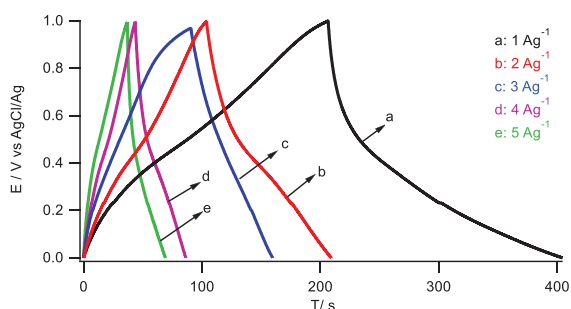


Fig 5. Galvanostatic charge/discharge curves of rGO in  $\text{H}_2\text{SO}_4$  1 M at different current densities

Table 2. Specific capacitance of rGO and rGO/PANI at different current densities

| $I (\text{A g}^{-1})$ | $C_{\text{rGO}} (\text{F g}^{-1})$ | $C_{\text{rGO/PANI}} (\text{F g}^{-1})$ |
|-----------------------|------------------------------------|-----------------------------------------|
| 5                     | 38                                 | 185,6                                   |
| 4                     | 42                                 | 307,3                                   |
| 3                     | 83                                 | 460,5                                   |
| 2                     | 115                                | 589                                     |
| 1                     | 182,7                              | 1121,2                                  |

The maximum specific capacitance value of the rGO/PANI is  $1121.2 \text{ F g}^{-1}$  at the current density of  $1 \text{ A g}^{-1}$ , and the minimum specific capacitance value is  $185.6 \text{ F g}^{-1}$  at  $I = 5 \text{ A g}^{-1}$ . These values are five times as high as the specific capacitance values at the responding current densities of rGO. This result is in consistence with previously published results of the conductive graphene-polymer material group known as the highest specific capacitance group up to now [5]. Obviously, the combination of PANI and graphene greatly improves the specific capacitance of material group, which is explained by the increase of surface area, boosting the electrical conductivity as well as improving the diffusion rate of ions during charge-discharge cycle [16]. This combination of PANI and graphene enhances the stability of the rGO/PANI structure during charge-discharge cycle, increasing the specific capacitance as well as the stability of the material [13].

Fig. 6 exhibits the results of the specific capacitance variation and the Coulombic efficiency of rGO and rGO/PANI in terms of current density. The Coulombic efficiency of rGO/PANI at different current density values is higher than that of rGO. This result shows that charge-discharge efficiency of rGO/PANI is better than that of rGO, which means that less electricity is consumed.

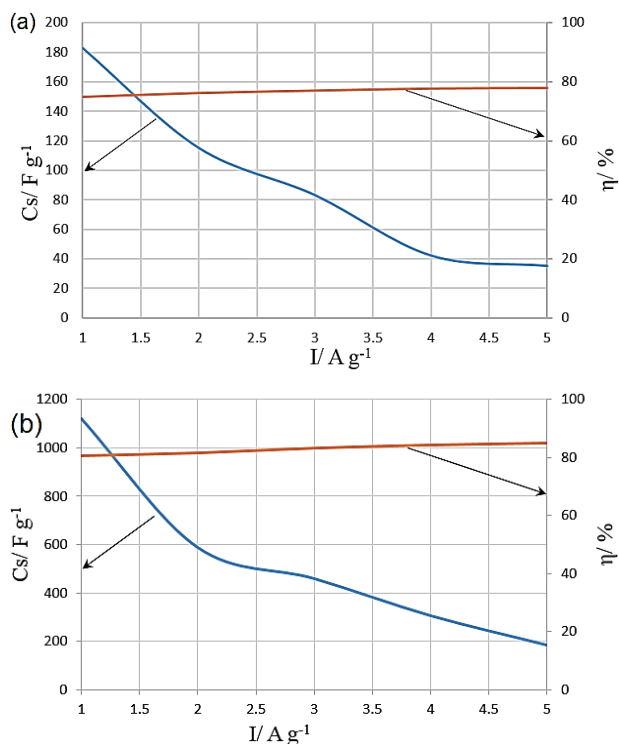
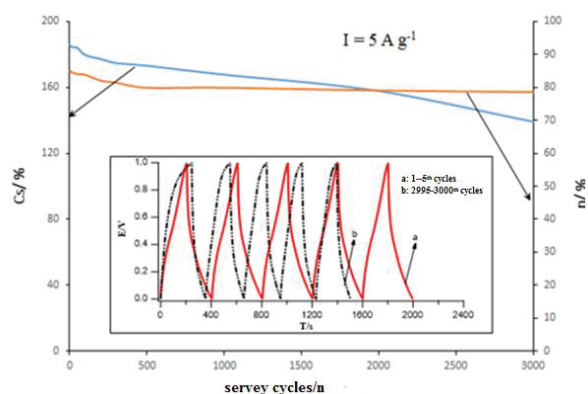


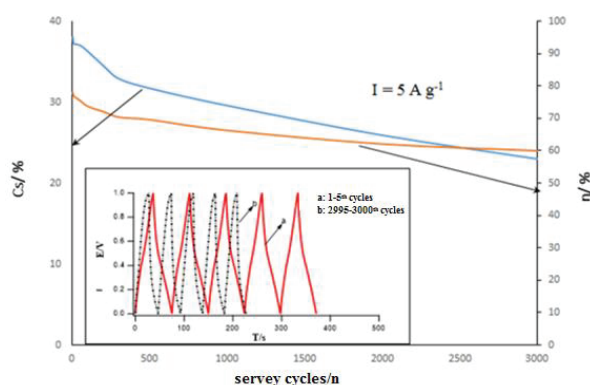
Fig 6. Specific capacitance and Coulombic efficient of rGO (a), rGO/PANI (b) in  $\text{H}_2\text{SO}_4$  1 M at different current densities.

#### Electrochemical stability of rGO/PANI

The result of determining the charge-discharge cycle, the specific capacitance variation and the Coulombic efficiency of rGO/PANI and rGO are respectively shown in Fig. 7, 8.



**Fig 7.** Specific capacitance and Coulombic efficient of rGO/PANI in  $\text{H}_2\text{SO}_4$  1 M at  $5 \text{ A g}^{-1}$  before 3000 cycles charge/discharge curves



**Fig 8.** Specific capacitance and Coulombic efficient of rGO in  $\text{H}_2\text{SO}_4$  1 M at  $5 \text{ A g}^{-1}$  before 3000 cycles charge/discharge curves

During 3000 of charge-discharge cycle, rGO/PANI showed the decrease in discharging time as well as its respective specific capacitance value. In early cycles, this decline was slow and unclear. As for rGO/PANI material, the first charge-discharge cycle earned the specific capacitance value of  $185.62 \text{ F g}^{-1}$ , and at the fifth cycle, the value was  $184.69 \text{ F g}^{-1}$ , down 0.5 % compared to the former. After 1000 cycles, the specific capacitance of rGO/PANI noticeably reduced as the discharging cycle decreased rapidly, and the charge-discharge cycle ratio became less than 1. At the 1000<sup>th</sup> cycle, the  $C_s$  of rGO/PANI was  $167.65 \text{ F g}^{-1}$ , down about 10 % compared to the first cycle. The specific capacitance values for this material at the 2000<sup>th</sup> and 3000<sup>th</sup> charge-discharge cycle were  $157.77 \text{ F g}^{-1}$  and  $139.22 \text{ F g}^{-1}$ , down 15 % and 25 % compared to the first cycle, respectively. At the same time, the rGO material showed a more significant reduce in charge-discharge cycle and specific capacitance values than rGO/PANI. Only after the first 5 charge-discharge cycle, the specific capacitance value of rGO decreased by 1.9 % compared to the original value; The decline at the 1000<sup>th</sup>, 2000<sup>th</sup> and 3000<sup>th</sup> charge-discharge cycle was 22.2 %, 30 % and 37.3 %, respectively, much higher than rGO/PANI.

The results obtained from rGO/PANI are better than those of Hang Sun's publication at the same conditions [6], when the rGO/PANI sample of this author has a 23 % reduction in the specific capacitance immediately after the first 1000 charge-discharge cycle

## Conclusions

A nanocomposite rGO/PANI, which can be directly used as electrodes material for a supercapacitor, was successfully prepared. The combination of PANI and graphene was significantly increased the surface area of the material from  $257.42 \text{ m}^2 \text{ g}^{-1}$  (for rGO) to  $478.26 \text{ m}^2 \text{ g}^{-1}$  (for rGO/PANI).

The specific capacitance of rGO/PANI in 1 M  $\text{H}_2\text{SO}_4$  solution is five times higher than that of rGO, at all currents density in the charge-discharge cycle measurement. The maximum specific capacitance of rGO/PANI is  $1121.2 \text{ F g}^{-1}$  compared to  $185.5 \text{ F g}^{-1}$  of rGO, at  $1 \text{ A g}^{-1}$

The electrochemical results also show the high activity stability of rGO/PANI in  $\text{H}_2\text{SO}_4$  1 M solution with the charge-discharge cycle measurement. Specifically, After 3000 cycles working at  $5 \text{ A g}^{-1}$ , the specific capacitance retention of rGO/PANI is about 75 % compared to the first cycle, higher than rGO (62.7 %).

The results show that the rGO/PANI material has the potential to be used in coating electrode of supercapacitors.

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